



This open access document is posted as a preprint in the Beilstein Archives at <https://doi.org/10.3762/bxiv.2020.139.v1> and is considered to be an early communication for feedback before peer review. Before citing this document, please check if a final, peer-reviewed version has been published.

This document is not formatted, has not undergone copyediting or typesetting, and may contain errors, unsubstantiated scientific claims or preliminary data.

Preprint Title The adsorption of 1-Chloro-1,2,2,2-tetrafluoroethane onto the pristine, Al-, and Ga-doped boron nitride nanosheet

Authors Mohsen Doust Mohammadi and Hewa Y. Abdullah

Publication Date 09 Dez. 2020

Article Type Full Research Paper

ORCID® iDs Mohsen Doust Mohammadi -
<https://orcid.org/0000-0003-2449-2962>; Hewa Y. Abdullah -
<https://orcid.org/0000-0001-5766-3858>

License and Terms: This document is copyright 2020 the Author(s); licensee Beilstein-Institut.

This is an open access work under the terms of the Creative Commons Attribution License (<https://creativecommons.org/licenses/by/4.0>). Please note that the reuse, redistribution and reproduction in particular requires that the author(s) and source are credited and that individual graphics may be subject to special legal provisions.

The license is subject to the Beilstein Archives terms and conditions: <https://www.beilstein-archives.org/xiv/terms>.

The definitive version of this work can be found at <https://doi.org/10.3762/bxiv.2020.139.v1>

The adsorption of 1-Chloro-1,2,2,2-tetrafluoroethane onto the pristine, Al-, and Ga-doped boron nitride nanosheet

Mohsen Doust Mohammadi^a, Hewa Y. Abdullah^{b,*}

^a School of Chemistry, College of Science, University of Tehran, Tehran 14176, Iran

^b Physics Education Department, Faculty of Education, Tishk International University, 44001, Erbil, Iraq

*Corresponding author Email: hewayaseen@gmail.com

Abstract

DFT were put into practice to study the nature of the intermolecular interactions between 1-Chloro-1,2,2,2-tetrafluoroethane (HCFC-124) gas molecule and pristine, aluminium, and gallium doped single-walled boron nitride nanosheets (BNNS). For performing optimization process, various functionals including PBE0, M06-2X, ω B97XD, and B3LYP-D3 were applied on both of the isolated and complex structures. All of the functionals were used together with split-valence triple-zeta basis sets with d-type Cartesian-Gaussian polarization functions (6-311G(d)). To consider the electronic structure, DOS analysis were employed. NBO, QTAIM, and NCI analyses were also taken on board to discover the nature of intermolecular interactions between gas and nanosheets using the same level of theory. The results of electronic structure calculations as well as population analyses has been carefully tabulated and partially depicted. The HOMO-LUMO energy gap was dramatically changed when the dopant atom added to the BNNS. It means the impurity can improve the sensivity and reactivity of the pristine nanosheet; therefore, by absorbing the HCFC-124 onto the surface of the titled nanosheets, a salient signal can produce in a typical electronic circuit. Among all of the absorbents, Al-doped BNNS shows the most favorable material to design a nanosensor for the studied gas molecule.

Keyword: Boron Nitride Nanosheet; 1-Chloro-1,2,2,2-tetrafluoroethane; HCFC-124; Freon 124; Natural Bond Orbital.

35 1. Introduction

36 After the emergence of density functional theory (DFT) due to its high computational power
37 for large molecules as well as its considerable accuracy and speed, a suitable framework was
38 provided for the theoretical studies of chemical systems. Many scientists today use this method to
39 understand the properties of chemical structures. Meanwhile, theoretical studies on nanoscale
40 structures have found an extraordinary place in computational chemistry, and researchers have
41 used numerous computational methods to study intermolecular interactions to design tools that
42 have better accuracy and performance [1-5]. A variety of nanomaterials have been considered for
43 the construction of sensors with high measurement accuracy, extreme insensitivity to temperature,
44 very low response time, low production costs, and resistant to harsh environmental conditions [6-
45 12].

46 Different of methods have been used in theoretical studies to improve the sensitivity of the
47 adsorption process in nanomaterials such as metal doping [13-15], surface defect [16-18],
48 transition metal decoration [19, 20], etc. Impurities introduced to the nanomaterial wall change the
49 energy gap, and also dramatically change its morphology. Lin et al. provides a comprehensive
50 study on the effect of transition metal decoration to boron nitride nanosheet (BNNS) [21]. Studies
51 have shown that even the introduction of non-metals such as oxygen can activate the surface of
52 BNNS [22]. Demonstrating the benefits of dopant element, we can point out that by reducing the
53 electrical resistance, impurity-decorated nanomaterials can generate a stronger signal in the circuit.
54 Therefore, compared to pure nanomaterials, it can be said that decorated nanomaterials have a
55 higher efficiency [23]. Aluminum and gallium are two elements that have been widely used to
56 activate the surface of boron nitride nanosheets in theoretical studies [24-27].

57 BNNSs are easily synthesized by the chemical vapor deposition method [28]. Due to its
58 remarkable mechanical, thermal, electrical and chemical properties, as well as its excellent
59 reactivity and selectivity, BNNS is widely used [29-31]. The wide band gap (5-6 eV) classified
60 BNNS as an electrical insulator [28, 32, 33]. Ahmadi et al. reported the values of energy gap and
61 B-N bond length for BNNS 5.93 eV and 1.45 Å, respectively, and for Al-doped BNNS the band
62 gap as well as Al-N bond length are 5.39 eV and 1.74 Å, respectively [34]BNNS is also a very
63 successful nanosorbent and has been studied in many theoretical studies to detect H₂O [35] NO
64 [36] , CO [37] , CH₂O [38], COCl₂ [39], C₂H₄ [40], and N₂O [41]. This nanosheet and metal
65 decorated BNNS have also been used in the manufacture of many electronic devices [42-48]. It
66 can also be used as a valuable material in the coating industry and metal protection because it is
67 impermeable to gases and liquids and is also an electrical insulator [49-52].

68 This article encompasses a comprehensive study to investigate the intermolecular
69 interactions of HCFC-124 gas molecule with pristine BNNS plus BN nanosheets doped with Al
70 and Ga. The details information about the computational level was given in section 2. The result
71 and discussion section has been divided in 4 separate subsections: Geometric Surveys (3.1.), which
72 provides insights about bond length and other geometry properties as well as introducing the levels
73 of study for optimizing all structures; Electronic Structure (3.2.) that discusses about DOS and
74 energetic properties results; NBO analysis (3.3.) and QTAIM analysis (3.4) are provided to show
75 the nature of intermolecular interactions between two fragments. We tried to provide a brief theory

76 for each part using useful references and avoided to bring boring description. And finally, the
77 conclusion section briefly refers to all the findings in this study.

78 2. Computational details

79 All of the nanosheet structures were optimized using two-dimensional periodic boundary
80 condition Kohn–Sham density functional theory [53-56] (PBC-DFT) method in vacuum. Various
81 functionals such as PBE0 [57-59], M06-2X [60] , ω B97XD [61] , and B3LYP-D3 scheme of
82 Grimme [62-64] together with split-valence triple-zeta basis sets with d-type Cartesian-Gaussian
83 polarization functions (6-311G(d)) [65-73] were employed through the spin-restricted approach.
84 There is no symmetry constraints were imposed. According to benchmark studies [74], such a
85 basis set covers all we need in this study. A $3 \times 3 \times 1$ k point sampling in the Brillouin zone
86 integration [75] with kinetic energy cutoff of 450 eV was hired. In this approach, when the number
87 of unit cells increase the total energy will become minimal. All the molecular geometries were
88 built in Gaussview 6.0.16 [76] software package then fully optimized by the Berny [77] method
89 by Gaussian 16 Rev. C.01 [78] linux based software. The optimization process were performed
90 using the default Gaussian convergence criteria. To configure the stability of each structure
91 frequency calculation were done to determine the nature of the stationary points. In addition, the
92 wave function stability calculation were also done to consider stability of the electrons. Natural
93 bond orbital (NBO) calculations were carried out using the NBO v 3.1 software, which is coupled
94 within Gaussian package to consider the charge-transfer interactions between occupied and
95 unoccupied orbitals. To extract the result data of NBO, NCI, and QTAIM analyses, Multiwfn [79]
96 software was used and GaussSum [80] package depicted the DOS diagrams.

97 The energy of adsorption (E_{ads}) between two fragments (nanosheet and HCFC-124) can be
98 considered as follows:

$$99 E_{\text{ads}} = E_{\text{sheet/HCFC-124}} - E_{\text{sheet}} - E_{\text{HCFC-124}} + \Delta E_{(\text{BSSE})} \quad (1)$$

100 where $E_{\text{sheet/HCFC-124}}$ shows the total energy of the gas/nanosheet cluster. E_{sheet} and $E_{\text{HCFC-124}}$ are the
101 energies of the isolated nanosheet and isolated gas molecule, respectively. Basis set superposition
102 error (BSSE) was obtained from the Boys and Bernardi's counterpoise procedure [81, 82] as
103 follows:

$$104 \Delta E_{(\text{BSSE})} = \Delta E_{\text{cluster}} - \Delta E_{\text{nanosheet}}^{\text{cluster}} - \Delta E_{\text{gas}}^{\text{cluster}} \quad (2)$$

105 According to Equation 1, negative values of E_{ads} (i.e. exothermic adsorptions) indicate that the
106 formed complex is stable; otherwise, positive values of E_{ads} is belong to a local minimum in which
107 the adsorption of HCFC-124 and nanosheet is deterred by a energy barrier.

108

109 3. Result and discussion

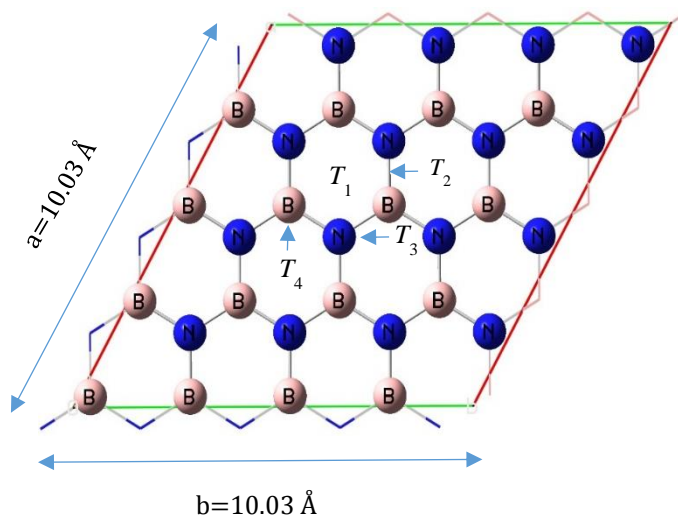
110 3.1 Geometric Surveys

111 The PBE0 functional, hybrid form of Perdew–Burke–Ernzerhof (PBE) [57], together with
112 6-311G(d) basis set were selected at the first stage to start the geometry optimization process of
113 the isolated and complex structures. Although PBE0 is fast, it cannot describe the long-range
114 interactions, dispersion effect, and charge transfer excitations; therefore, all of the structures were
115 re-optimized using robust methods like M06-2X, ω B97XD, and B3LYP-D3. Head-Gordon et al.
116 [61] invented the ω B97XD functional to account for the effect of long-range interactions as well
117 as the dispersion corrections. Among Minnesota 06 suit, which is developed by Truhlar et al. [83]

118 , the global hybrid functional with double non-local exchange (2X) amounts (i.e. M06-2X) is a
119 high performance method to study the non-covalent interactions. To study the dispersion effect,
120 the latest version of B3LYP-D3 known as D3 (BD) (GD3BJ) developed by Grimme et al. [62-64]
121 would be a good option. In order to compare the effect of each of the mentioned corrections, in
122 Table 1, the amounts of adsorption energy obtained from PBE0 method can be evaluated with
123 other more complex aforementioned methods.

124 The PBC-DFT framework at PBE0/6-311G(d) level was applied on a pristine $B_{18}N_{18}$ unit
125 cell with 10.004 \AA in each dimension. Figure 1 shows the selected boron nitride nanosheet
126 (BNNS). Then the unit cell expanded 3 times along with each axis and the optimization process
127 repeated with the same level of theory. It should be noted that, to reduce the boundary effects, the
128 terminal atoms were terminated by H atoms. For the case for Al and Ga doped nanosheets, after
129 the optimized expanded BNNS obtained, one of the boron atoms substituted by Al or Ga elements
130 and calculations were performed at PBE0/6-311G(d) level of theory. The re-optimization process
131 were performed using 3 other functionals and same basis set. The values of bond length for BNNS,
132 BNAlNS, and BNGaNS are depicted in Figure 2. As is obvious from Figure 2 (a), the B-N bond
133 length is 1.45 \AA . Also, Figures 2 (b) and (c) show the value of bond length for Al-N and Ga-N
134 1.71 \AA and 1.74 \AA , respectively. Due to the inner stress, the B-N bond length in the vicinity
135 of Al ad Ga dopant atoms vary from 1.42 \AA to 1.49 \AA .

136



137

138 **Figure 1:** The boron nitride unit cell (32 atoms, 192 electrons, neutral, and singlet) prepared to
139 perform two-dimensional periodic boundary condition density functional theory calculations.
140

141

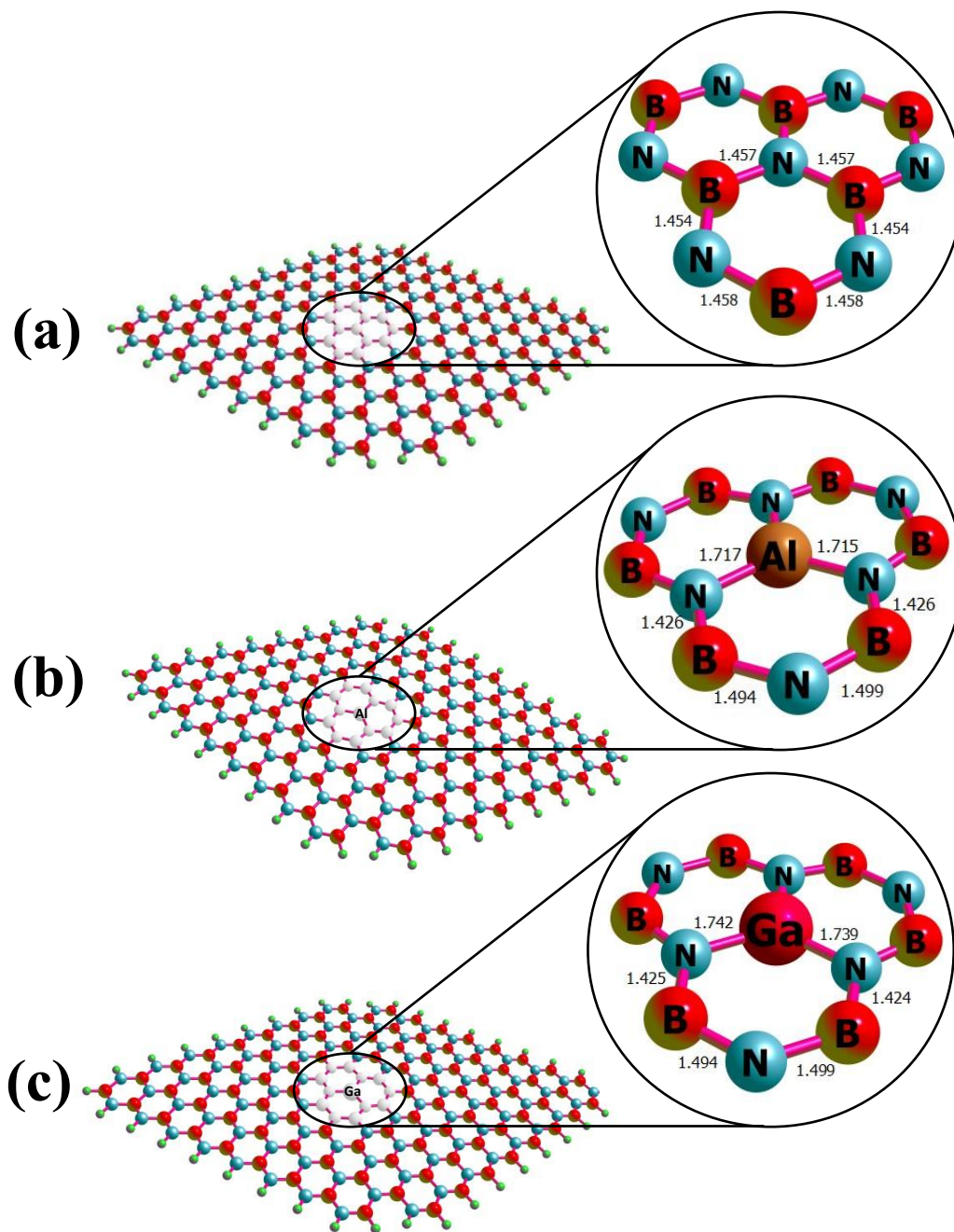
142

143

144

145

146
147
148
149



150
151

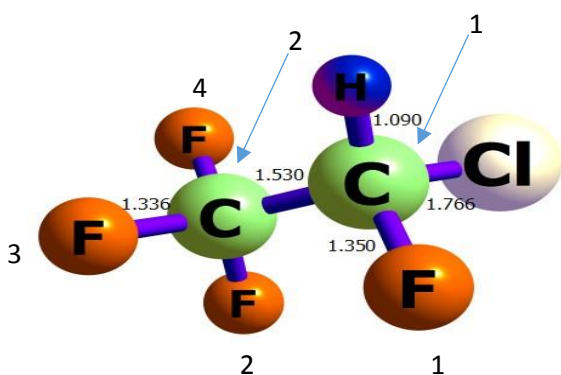
152 **Figure 2:** The bond length (Å) obtained at B3LYP-D3/6-311(d) level for (a) BNNS, (b) BNAINS,
153 and (c) BNGaNS.

154
155

156 The adsorption of HCFC-124 gas molecule onto the surface of pristine and Al, Ga-doped BNNS
157 were also carried out. To achieve this goal, the gas molecule should be placed on top of the
158 nanosheet in different positions through different orientations. The purpose of such calculations is
159 to find energy local minima. Figure 3 shows the different heads of HCFC-124 including H, Cl and
160 different F heads. Carbon atoms are also numbered to be separable in the coming tables. And there
161 are 4 different sites in the BNNS (Figure 1) including: T₁, in the middle of hexagonal ring; T₂,
162 between B and N atoms; T₃, N atom; and T₄, B atom. The gas molecule were put on top of each
163 T_x sites through different distances and optimization process started at PM7/6-311G(d) level. It
164 should be stressed that, overall, there are 240 initial orientations were prepared (i.e. 6×4×10=240.
165 It means 6 different gas molecule heads were put on each of the 4 sites of nanosheet from 0.5 to
166 5.0 Å vertical distances with 0.5 Å intervals). Right after that, 12 minima were predicted. We select
167 these 12 complex structures to optimize using PBE0/6-311G(d) level of theory. Regarding the
168 values of adsorption energies, finally, the most stable complexes were chosen and re-optimization
169 were done using other functionals including M06-2X, ωB97XD, and B3LYP-D3. The relaxed
170 structures of all complexes which has been obtained from B3LYP-D3/6-311G(d) are depicted in
171 Figure 4. When the values of E_{ads} are “below the range of chemical interest” [84], such results
172 provides identical results.

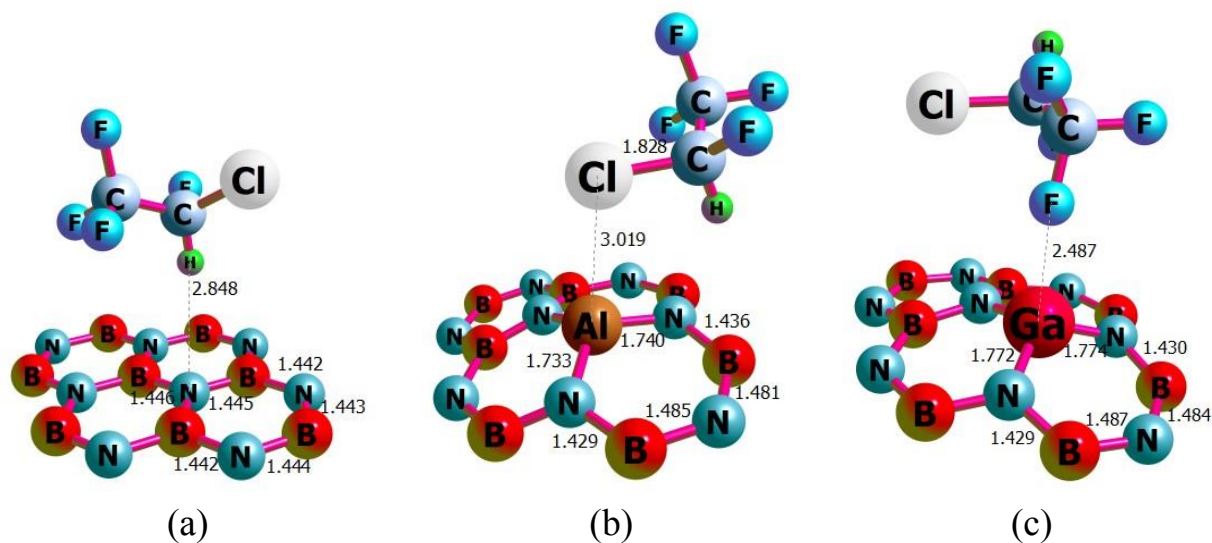
173 The adsorption energy obtained from ωB97XD and B3LYP-D3 are closed to each other. The
174 ωB97XD functional shows the most negative E_{ads} values according to Table 1. It seems these two
175 functional are good options for further investigations in this study. The most negative value of
176 E_{ads} is corresponds to the adsorption of gas onto the Al-doped BNNS, -1.586 eV which indicates
177 that the reactivity of HCFC-124 gas is significant. Otherwise, the reactivity of gas and pristine
178 BNNS shows the weakest interaction, among those doped sheets. We preferred to run the
179 population analysis calculations with B3LYP-D3 to reduce the computation times. Therefore, all
180 the interactions analyses were applied to the results of the B3LYP-D3/6-311G(d) level of theory.

181



182
183 **Figure 3:** The relaxed structure of 1-Chloro-1,2,2,2-tetrafluoroethane gas molecule obtained from
184 B3LYP-D3/6-311G(d) level of theory.
185

186
187
188
189
190
191
192
193



194
195
196
197
198
199
200
201
202
203
204

Figure 4: The relaxed structures of a) HCFC-124/BNNS, b) HCFC-124/BNAINS, and c) HCFC-124/BNGaNS obtained from B3LYP-D3/6-311G(d) level of theory.

Table1: The adsorption energies (E_{ads}) HCFC-124/nanosheet complexes. All values are in eV and obtained from geometry optimization calculations using PBE0, M06-2X, ω B97XD, and B3LYP-D3 functional in combination with 6-311G(d) basis set.

Systems	PBE0	B3LYP-D3	M06-2X	ω B97XD
C_2HCIF_4 /BNNS	-0.767	-0.899	-0.776	-1.064
C_2HCIF_4 /BNAINS	-1.117	-1.438	-1.126	-1.586
C_2HCIF_4 /BNGaNS	-1.057	-1.416	-1.207	-1.518

205
206
207
208
209

210
211
212
213

3.2 Electronic Structure

214 The “Conceptual DFT” has been developed to consider the reactivity concept. Various
215 properties can be obtained from HOMO-LUMO energy gap (HLG) [85] in such a way when an
216 external potential is applied to a system, in the Hohenberg–Kohn theorems [53] context,
217 changes as follows:

$$\begin{aligned}
 \Delta E &\equiv E[N + \Delta N, v(\mathbf{r}) + \Delta v(\mathbf{r})] - E[N, v(\mathbf{r})] \\
 &= \left(\frac{\partial E}{\partial N} \right)_v \Delta N + \int \left(\frac{\delta E}{\delta v(\mathbf{r})} \right)_N \delta v(\mathbf{r}) d(\mathbf{r}) + \\
 &\quad \frac{1}{2!} \left\{ \left(\frac{\partial^2 E}{\partial N^2} \right)_v \Delta N^2 + 2 \int \left(\frac{\partial}{\partial N} \left(\frac{\delta E}{\delta v(\mathbf{r})} \right) \right)_N \Delta N \delta v(\mathbf{r}) + \iint \left(\frac{\partial^2 E}{\partial v^2(\mathbf{r})} \right)_N \delta v(\mathbf{r}) \delta v(\mathbf{r}') d\mathbf{r} d\mathbf{r}' \right\} + \dots
 \end{aligned} \tag{3}$$

219 In the above Taylor expansion N is a global quantity and $v(\mathbf{r})$ is a local function. Each term has a
220 specific meaning in the chemical language as follows:

$$-\chi = \left(\frac{\partial E}{\partial N} \right)_v = \mu \cong \frac{(\varepsilon_{LUMO} + \varepsilon_{HOMO})}{2} \tag{4}$$

$$\eta = \left(\frac{\partial \mu}{\partial N} \right) = \frac{1}{2} \left(\frac{\partial^2 E}{\partial N^2} \right)_v = \frac{1}{2} (IP - EA) \tag{5}$$

$$\omega = \frac{\mu^2}{2\eta} \tag{6}$$

224 Equation 4 shows the negative electronegativity (χ) which is equal to chemical potential (μ). Also
225 the values of HOMO and LUMO are related to the ionization affinity and electron affinity,
226 respectively. You can follow the other terms to reach to the Fukui function, response function,
227 dual descriptor, etc [86, 87]. And Equation 6 is related to the electrophilicity index (ω). The values
228 of these properties are listed in Table 2.

229 The energy gap (E_g) of BNNS has been calculated about 5.78 eV at B3LYP-D3/6-311G (d) level
230 of theory and the adsorption of HCFC-124 on it reduced the energy gap to 5.72 which shows
231 although adsorption energy is about -0.9 eV the sensitivity is not significant. On the other hand,
232 Al- and Ga-doped nanosheets meaningfully reduced the E_g values. Table 2 Shows the LUMO
233 values were largely stabilized then the E_g has been reduced. For the case of BNAINS this reduction
234 of E_g is higher than BNGaNS; therefore, the sensor response (S) is more stronger according to the
235 following equations:

$$S = \left| \left(\frac{\sigma_1}{\sigma_2} \right) - 1 \right| = \exp \left(\frac{|\Delta E_g|}{kT} \right) - 1 \tag{7}$$

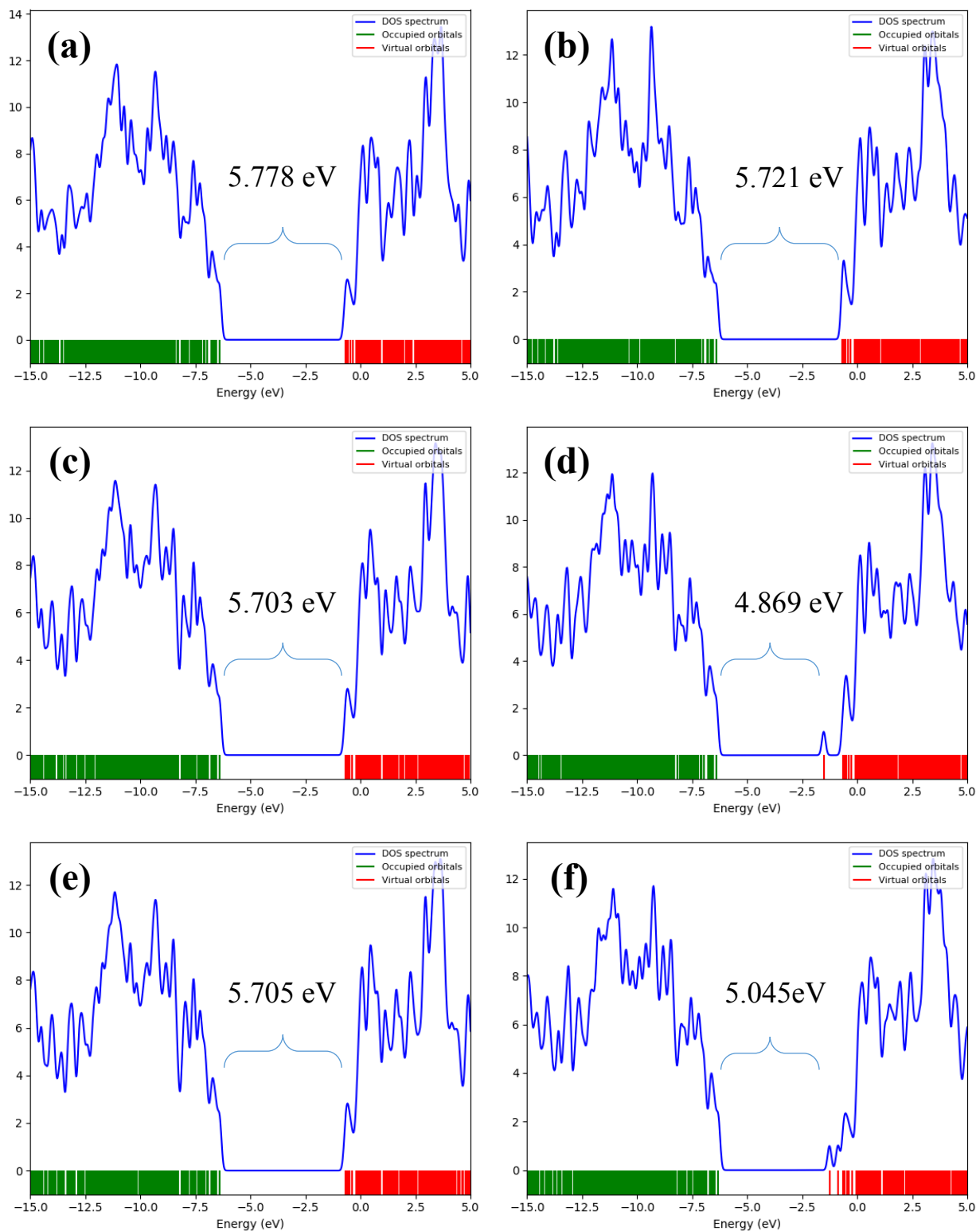
237
$$\sigma = AT^{3/2}e^{\frac{E_g}{2kT}}$$
 (8)

238 Where A is a constant and σ is electrical conductivity, k is Boltzmann's constant, and T is Kelvin
 239 temperature. The resistivity will be diminished when E_g is being reduced, since the resistivity is
 240 proportional to the reciprocal of the conducting electron population. Hence, the resistivity for
 241 BNAINS/gas adsorption is low and the electric current generated in the circuit will face the lowest
 242 resistance. Density of state (DOS) map is useful in intuitively revealing density of distribution of
 243 molecular orbitals in different energy regions, and gap is directly visible from this map (Figure 5).

244 **Table 2:** HOMO energy (ϵ_H), LUMO energy (ϵ_L), Fermi energy (E_F), HOMO–LUMO energy gap
 245 (HLG), chemical potential (μ), chemical hardness (η), and electrophilicity (ω). All values are in
 246 eV and were obtained using the B3LYP-D3/6-311G (d) level of theory.

Systems	ϵ_H	ϵ_L	E_F	HLG	μ	η	ω
BNNS	-6.474	-0.695	-3.584	5.778	-3.584	2.889	18.560
BNAINS	-6.389	-0.686	-3.538	5.703	-3.538	2.851	17.841
BNGaNS	-6.391	-0.687	-3.539	5.705	-3.539	2.852	17.862
C₂HCIF₄/BNNS	-6.386	-0.664	-3.525	5.721	-3.525	2.861	17.772
C₂HCIF₄/BNAINS	-6.389	-1.520	-3.955	4.869	-3.955	2.435	19.037
C₂HCIF₄/BNGaNS	-6.305	-1.260	-3.782	5.045	-3.782	2.523	18.043

247



248
 249 **Figure 5:** Density of state maps for (a) BNNS (b) HCFC-124/BNNS (c) BNAINS (d) HCFC-124/BNAINS
 250 (e) BNGaNS (f) HCFC-124/BNGaNS. Data were obtained from the B3LYP-D3/6-311G (d) level of theory
 251

3.3 NBO Analysis

The Natural bond orbital (NBO) method, developed by Weinhold et al. [88-90], is one of the most respectful population analyses and uses to calculate the distribution of electron density in bonds between atoms. The term NBO refers to a bonding orbital with the maximum electron density. A density matrix, calculated from DFT, as well as atomic charge, are used to define natural bonding orbitals. To complete the span of valance space in addition to bonding NBO (σ), we need an antibonding NBO (σ^*) as follows:

$$\sigma_{AB} = C_A h_A + C_B h_B \quad (9)$$

$$\sigma_{AB}^* = C_A h_A - C_B h_B \quad (10)$$

where h_A and h_B are natural hybrid valance orbitals, C_A and C_B are the corresponding polarization coefficients. In the present study, NBO calculations were performed to figure out various types of bond order including Mulliken [91](Equation 11) and Mayerb [92-94] (Equation 12) bond order as well as Wiberg bond index (WBI) in Löwdin orthogonalized basis [95, 96] (Equation 13). Thus,

$$I_{AB} = \sum_i \eta_i \sum_{a \in A} \sum_{b \in B} 2C_{a,i} C_{b,i} S_{a,b} = 2 \sum_{a \in A} \sum_{b \in B} P_{a,b} S_{a,b} \quad (11)$$

$$I_{AB} = I_{AB}^\alpha + I_{AB}^\beta = 2 \sum_{a \in A} \sum_{b \in B} \left[(P^\alpha S)_{ba} (P^\alpha S)_{ab} + (P^\beta S)_{ba} (P^\beta S)_{ab} \right] \quad (12)$$

$$I_{AB} = \sum_{a \in A} \sum_{b \in B} P_{ab}^2 \quad (13)$$

In the above Equations, P and S are density and overlap matrix, respectively. Mulliken and Mayer's bond orders are sensitive to the basis set, spacially for the basis set including diffuse functions. On the other hand, Wiberg bond order with respect to the two other is less basis set dependence. Table 3 reports the values of obtained bond orders from different methods. According to the WBI which is more accurate than Mayer and Mulliken we can up to this conclusion that BNAINS adsorbent is the most material in this study for adsorbing HCFC-124. The bond order value shows that the interaction of gas with BNNS can be classified as physisorption; otherwise, the interactions between Al and Ga-doped BNNS with the gas molecules are more significant.

In addition, natural charges and natural electron configuration data were obtained. Using NBO consideration the nature of intermolecular interaction between HCFC-124 gas molecule and nanosheets becomes more understandable. According to Table 4, the charge transfer is obvious. The valance shell of F atom was changed from 2p(5.50) to 2p(5.49)3d(0.01) during the adsorption process. It means strong interaction has occurred.

Table 3: The values of Mulliken, Mayer, and Wiberg bond order obtained from the interactions of the HCFC-124 molecule and BNNS, BNAINS, and BNGaNS. All calculations were performed using the B3LYP-D3/6-311G (d) level of theory.

Systems	X ^a ...Y ^b	Mulliken	Mayer	WBI
C₂HCIF₄/BNNS	H....N	0.290	0.022	0.005
C₂HCIF₄/BNAINS	Cl....Al	0.060	0.115	0.254
C₂HCIF₄/BNGaNS	F....Ga	0.102	0.124	0.244

291 (a) X atoms are belong to HCFC-124

292 (b) Y atoms are belong to Nanosheets

293

294

295

296 **Table 4:** The values of natural charges (au) and Natural electron configurations for the isolated
 297 HCFC-124, and its complex structures. All values were calculated using the B3LYP-D3/6-311G
 298 (d) level of theory.

Systems	atom	Natural Charge	Natural Electron Configuration
C₂HCIF₄	C1	1.06	[core]2S(1.03)2p(2.76)3S(0.01)3p(0.02)3d(0.01)
	C2	0.16	[core]2S(0.76)2p(2.11)3p(0.04)3d(0.02)
	H	0.18	1S(0.81)
	Cl	-0.02	[core]3S(1.86)3p(5.15)3d(0.01)4p(0.01)
	F1	-0.34	[core]2S(1.84)2p(5.50)
	F2	-0.35	[core]2S(1.84)2p(5.50)
	F3	-0.34	[core]2S(1.84)2p(5.49)
	F4	-0.35	[core]2S(1.84)2p(5.50)
C₂HCIF₄/BNNS	C1	1.06	[core]2S(0.78)2p(2.08)3p(0.03)3d(0.02)
	C2	0.18	[core]2S(1.03)2p(2.76)3S(0.01)3p(0.02)3d(0.01)
	H	0.19	1S(0.80)
	Cl	-0.04	[core]3S(1.88)3p(5.12)3d(0.01)4p(0.01)
	F1	-0.35	[core]2S(1.85)2p(5.49)3d(0.01)
	F2	-0.35	[core]2S(1.85)2p(5.49)3d(0.01)
	F3	-0.35	[core]2S(1.85)2p(5.49)3d(0.01)
	F4	-0.35	[core]2S(1.85)2p(5.49)3d(0.01)
C₂HCIF₄/BNAINS	C1	1.07	[core]2S(0.78)2p(2.08)3p(0.03)3d(0.02)
	C2	0.17	[core]2S(1.03)2p(2.76)3S(0.01)3p(0.02)3d(0.01)
	H	0.21	1S(0.80)
	Cl	-0.01	[core]3S(1.88)3p(5.12)3d(0.01)4p(0.01)
	F1	-0.34	[core]2S(1.85)2p(5.49)3d(0.01)
	F2	-0.34	[core]2S(1.85)2p(5.49)3d(0.01)
	F3	-0.35	[core]2S(1.85)2p(5.49)3d(0.01)
	F4	-0.36	[core]2S(1.85)2p(5.49)3d(0.01)
C₂HCIF₄/BNGaNS	C1	1.07	[core]2S(0.78)2p(2.08)3p(0.03)3d(0.02)
	C2	0.17	[core]2S(1.03)2p(2.76)3S(0.01)3p(0.02)3d(0.01)
	H	0.20	1S(0.80)

C1	-0.01	[core]3S(1.88)3p(5.12)3d(0.01)4p(0.01)
F1	-0.34	[core]2S(1.85)2p(5.49)3d(0.01)
F2	-0.34	[core]2S(1.85)2p(5.49)3d(0.01)
F3	-0.35	[core]2S(1.85)2p(5.49)3d(0.01)
F4	-0.34	[core]2S(1.85)2p(5.49)3d(0.01)

299

300

301 3.4 QTAIM Analysis

302 QTAIM analysis is used study bond types and intermolecular interactions. A critical point
303 of the electron density, including minimum, maximum, or saddle point, can belong to: (1) *Atomic*
304 *critical point* (ACP); (2) *bond critical point* (BCP); (3) *ring critical point* (RCP); and (4) *cage*
305 *critical point* (CCP). The Electron density $\rho(\mathbf{r})$ and the Laplacian electron density $\nabla^2\rho(\mathbf{r})$ are
306 playing important role in the QTAIM analysis since they determine the segmentation and
307 identification of different types of chemical interactions. A bond critical point with negative values
308 of Laplacian electron density and large values of electron density ($\rho(\mathbf{r}) > 10^{-1}$ a.u.) is defined as a
309 covalent bond. On the other hand, the positive value of $\nabla^2\rho(\mathbf{r})$ designates that the interactions can
310 be classified as of the non-substrate close-shell type (which include ionic and van der Waals
311 interactions) [97].

312 The values of Lagrangian kinetic energy $G(\mathbf{r})$ and potential energy density $V(\mathbf{r})$ divulges the
313 nature of the intermolecular interaction; therefore, the ratio of $G(\mathbf{r})/|V(\mathbf{r})|$ can be hired as an
314 appropriate index in link classification. When $G(\mathbf{r})/|V(\mathbf{r})| < 0.5$, the nature of the interaction is
315 covalent, and if $G(\mathbf{r})/|V(\mathbf{r})| > 1$, the interaction is non-covalent. Large values of elliptical bond (ε)
316 represents an unstable structure and defined as follows [98] :

$$317 \quad \varepsilon = \frac{\lambda_1}{\lambda_2} - 1, |\lambda_1| > |\lambda_2| \quad (14)$$

318 According to virial theorem [99], a relationship exists between $G(\mathbf{r})$, $V(\mathbf{r})$, and $\nabla^2\rho(\mathbf{r})$ as
319 follows:

$$320 \quad \frac{1}{4}\nabla^2\rho(\mathbf{r}) = 2G(\mathbf{r}) + V(\mathbf{r}) \quad (15)$$

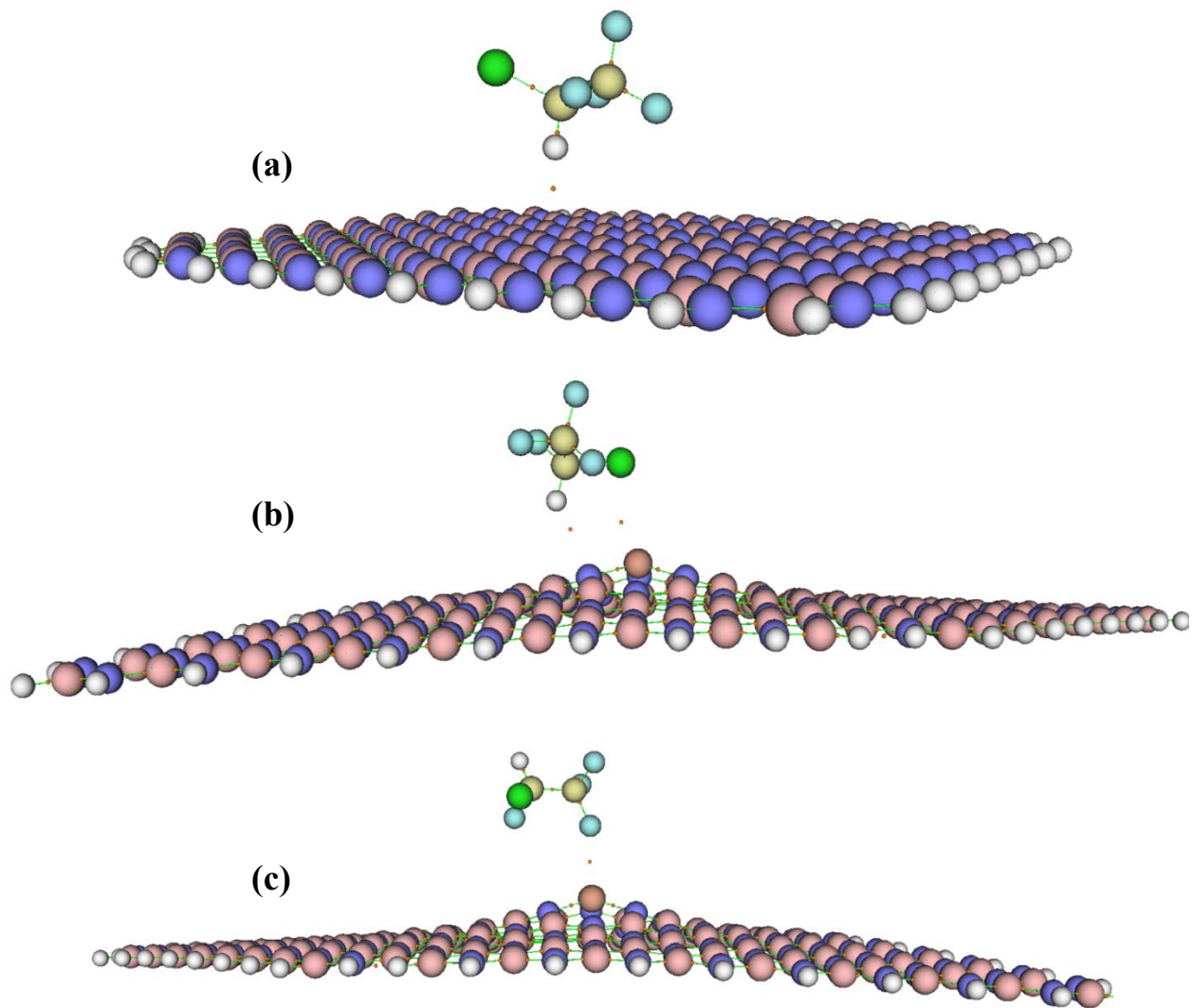
321 From Table 5 all values for $\nabla^2\rho(\mathbf{r})$ are positive. except Cl...Al interaction between HCFC-
322 124 and BNAINS, the other values of $G(\mathbf{r})/|V(\mathbf{r})|$ are higher than 1. These results shows
323 intermolecular interactions are classified as non-covalent. In the case of BNAINS very strong
324 interactions were observed. And finally the small values of ε shows that all of the structures are
325 stable.

326

327

328

329



330

331

332 **Figure 6:** The Bond Critical points graphs for the (a) HCFC-124/BNNS, (b) HCFC-124/BNAINS,
 333 (c) and HCFC-124/BNGaNS systems. The orange dots represent the BCPs

334

335

336

337

338

339

340

341

342

343

344

345

346

347 **Table 5:** QTAIM topological parameters for electron density $\rho(\mathbf{r})$, Laplacian of electron density
 348 $\nabla^2\rho(\mathbf{r})$, kinetic electron density $G(\mathbf{r})$, potential electron density $V(\mathbf{r})$, eigenvalues of Hessian matrix
 349 (λ), and bond ellipticity index (ε) at the BCPs of the HCFC-124 clusters with BNNS, BNAINS,
 350 and BNGaNS. All values were calculated using the B3LYP-D3/6-311G (d) level of theory and
 351 NBO analysis.

Systems	Bond	ρ	∇^2r	$G(\mathbf{r})$	$V(\mathbf{r})$	$G(\mathbf{r})/V(\mathbf{r})$	λ_1	λ_2	λ_3	ε
$\text{C}_2\text{HCIF}_4/\text{BNNS}$	H...B	0.0057	0.0179	0.0037	-0.0029	1.2781	-0.0021	-0.0038	0.0238	0.7598
$\text{C}_2\text{HCIF}_4/\text{BNAINS}$	H...N	0.0121	0.0371	0.0079	-0.0066	1.2076	-0.0100	-0.0105	0.0577	0.0501
	Cl...Al	0.0121	0.0195	0.0053	-0.0058	0.9227	-0.0051	-0.0078	0.0324	0.5138
$\text{C}_2\text{HCIF}_4/\text{BNGaNS}$	F...Ga	0.0193	0.0796	0.0195	-0.0190	1.0220	-0.0163	-0.0171	0.1130	0.0504

352

353

354 The results of QTAIM, in the previous section, showed that the interactions between HCFC-
 355 124 and nanosheets are non-covalent; hence, it is useful to check them by a non-covalent analysis.
 356 Reduced density gradient (RDG) and $\text{sign}\lambda_2(r)\rho(r)$ are a pair of functions used in non-covalent
 357 interaction (NCI) [100] analysis which can be implemented to visualize the region and the type of
 358 weak interactions. RDG is defined as follows [100, 101]:

359

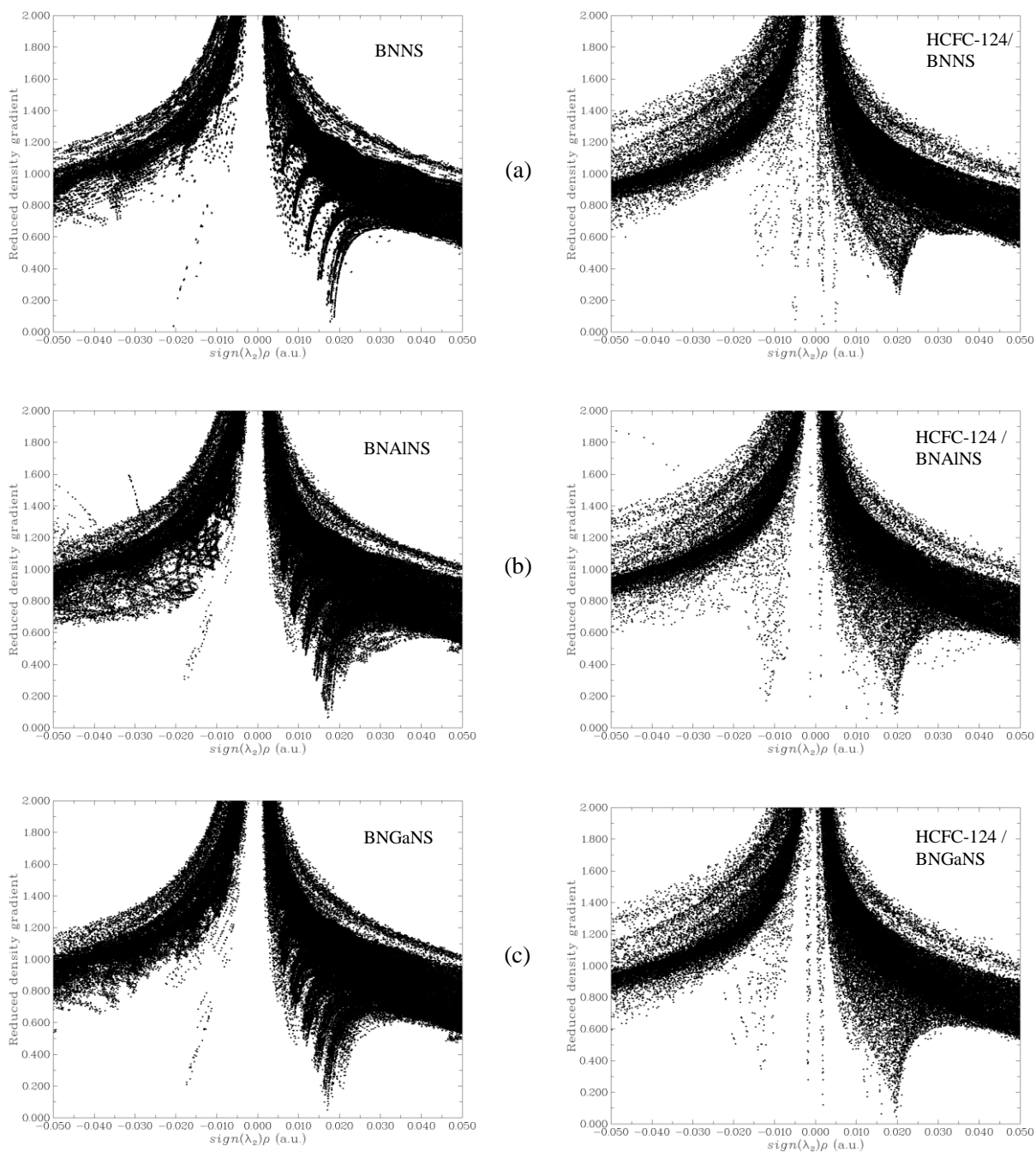
$$360 \quad \text{RDG} = \frac{1}{2(3\pi^2)^{\frac{1}{3}}} \frac{|\overline{\Delta\rho(\mathbf{r})}|}{\overline{\rho(\mathbf{r})}^{\frac{4}{3}}} \quad (16)$$

361

362 The two functions RDG and $\text{sign}\lambda_2(\mathbf{r})\rho(\mathbf{r})$ can be plotted to define specific areas. In this
 363 case, non-covalent interactions will be identified. The points that indicate strong interactions are
 364 located in the $\text{sign}\lambda_2(\mathbf{r})\rho(\mathbf{r}) < 0$ region. Relatively weak van der Waals interactions are found in
 365 the $\text{sign}\lambda_2(\mathbf{r})\rho(\mathbf{r}) \approx 0$ region. And if points are in $\text{sign}\lambda_2(\mathbf{r})\rho(\mathbf{r}) > 0$ region, it means that the
 366 interactions are of the type of repulsive [100, 101]. As it turns out, bond strength is closely related
 367 to the density matrix $\rho(\mathbf{r})$ as well as $\text{sign}\lambda_2$. Low RDG and low electron density regions should be
 368 consulted to determine whether non-covalent interactions occur between the two components
 369 involved in the adsorption process.

370

371 Considering an isosurface as a reference (e.g. RDG = 0.5), it can be seen that the spots are
 372 appeared in the $\text{sign}\lambda_2(r)\rho(r) \approx 0$ zone after the adsorption of HCFC-124 molecule onto BNNS
 373 and BNGaNS; therefore, the interactions can be classified as van der Waals. However, the
 374 interactions of HCFC-124 with BNAINS were strong in nature. NCI analysis also confirms the
 375 results of the adsorption energy calculations, QTAIM analysis, and NBO analysis, namely that the
 376 interactions of HCFC-124 with the Al and Ga-doped BNNS were strong.



377
 378 **Figure 7:** The RDG vs. $\text{sign}(\lambda_2)\rho(r)$ diagrams for (a) pristine (b) Al-doped, and (c) Ga-doped
 379 BNNs. The diagrams were obtained from B3LYP-D3/6-311G(d) level of theory. Left panel
 380 shows isolated nanosheets and right panel shows HCFC-124/nanosheet complexes.

381
 382
 383
 384

385 **4. Conclusion**

386 The intermolecular interactions between the HCFC-124 gas molecule and BNNS, BNAINS,
387 and BNGaNS were studied by the DFT framework in vacuum condition. All molecular structures
388 optimized at PBE0, M06-2X, ω B97XD, and B3LYP-D3 functionals together with a 6-311G(d)
389 basis set. Relaxed structures obtained from B3LYP-D3/6-311G(d) were chosen for population
390 analysis calculations. Results of adsorption energy show that among the nanosheets, the interaction
391 of BNAINS and gas (about -1.59 eV) is higher than the other adsorbents. DOS analysis can also
392 approve that the most reduction of energy gap (about 0.83 eV) is related to the gas/BNAINS
393 cluster. Different bond order analysis data repeats former results. To shed light on the nature of
394 intermolecular interactions NBO QTAIM, and NCI analyses were implemented and the results of
395 all the analyses were in agreement. From NBO analysis, the charge transfer was observed and NCI
396 and QTAIM results show strong intermolecular interactions-specially for BNAINS/gas cluster. To
397 sum up, we can conclude injecting Al and Ga elements inside the BNNS can dramatically active
398 its surface in favor of adsorbing HCFC-124 gas. Accordingly, these nanomaterials would be
399 favorable for designing a sensitive nanosensor.

400 **Conflict of Interest**

401 The authors declare no conflict of interest

402

403 **Acknowledgements**

404 I would like to thank the Solid-State Theory Group at the Physics Department at the Universita' degli Studi
405 di Milano-Italy for providing computational facilities.

406

407

408

409

410

411

412

413

414

415

416

417

418

419

420

421

422

423

424 **References**

- 425 [1] R. Dos Santos, R. Rivelino, F. de Brito Mota, G. Gueorguiev, and A. Kakanakova-Georgieva. *Journal*
426 *of Physics D: Applied Physics*. **2015**, 48, (29), 295104.
- 427 [2] M. D. Mohammadi, and M. Hamzehloo. *Computational and Theoretical Chemistry*. **2018**, 1144,
428 26-37.
- 429 [3] H. Ghafur Rauf, S. Majedi, E. Abdulkareem Mahmood, and M. Sofi. *Chemical Review and Letters*.
430 **2019**, 2, (3), 140-150.
- 431 [4] M. Kamel, A. Morsali, H. Raissi, and K. Mohammadifard. *Chemical Review and Letters*. **2020**, 3,
432 (1), 23-37.
- 433 [5] F. Li, and H. Asadi. *Journal of Molecular Liquids*. **2020**, 113139.
- 434 [6] E. Nemati-Kande, M. Abbasi, and M. Doust Mohammadi. *ChemistrySelect*. **2018**, 3, (34), 9833-
435 9840.
- 436 [7] E. Nemati-Kande, M. Abbasi, and M. D. Mohammadi. *ChemistrySelect*. **2019**, 4, (8), 2453-2462.
- 437 [8] M. D. Mohammadi, and H. Y. Abdullah. *Journal of Molecular Modeling*. **2020**, 26, (10), 1-15.
- 438 [9] M. D. Mohammadi, and H. Y. Abdullah. *Theoretical Chemistry Accounts*. **2020**, 139, (10), 1-17.
- 439 [10] M. D. Mohammadi, and H. Y. Abdullah. *Structural Chemistry*. **2020**, 1-14.
- 440 [11] M. D. Mohammadi, I. H. Salih, and H. Y. Abdullah. *Journal of Computational Biophysics and*
441 *Chemistry*. **2020**, 1-17.
- 442 [12] E. Nemati-Kande, M. Abbasi, and M. D. Mohammadi. *Journal of Molecular Structure*. **2020**, 1199,
443 126962.
- 444 [13] K. Wilke, and H. Breuer. *Journal of Photochemistry and Photobiology A: Chemistry*. **1999**, 121, (1),
445 49-53.
- 446 [14] D. Saha, and S. Deng. *Langmuir*. **2009**, 25, (21), 12550-12560.
- 447 [15] J. A. Botas, G. Calleja, M. Sánchez-Sánchez, and M. G. Orcajo. *Langmuir*. **2010**, 26, (8), 5300-5303.
- 448 [16] L.-Q. Wang, D. R. Baer, M. H. Engelhard, and A. N. Shultz. *Surface science*. **1995**, 344, (3), 237-
449 250.
- 450 [17] K. Bolton. *Journal of Molecular Structure: THEOCHEM*. **2003**, 632, (1-3), 145-156.
- 451 [18] Z. Wu, M. Li, J. Howe, H. M. Meyer III, and S. H. Overbury. *Langmuir*. **2010**, 26, (21), 16595-16606.
- 452 [19] T. Yildirim, J. Íñiguez, and S. Ciraci. *Physical Review B*. **2005**, 72, (15), 153403.
- 453 [20] K. Srinivasu, and S. K. Ghosh. *The Journal of Physical Chemistry C*. **2012**, 116, (48), 25184-25189.
- 454 [21] S. Lin, X. Ye, R. S. Johnson, and H. Guo. *The Journal of Physical Chemistry C*. **2013**, 117, (33), 17319-
455 17326.
- 456 [22] W. Lei, H. Zhang, Y. Wu, B. Zhang, D. Liu, S. Qin, Z. Liu, L. Liu, Y. Ma, and Y. Chen. *Nano Energy*.
457 **2014**, 6, 219-224.
- 458 [23] M. Hjiri, L. El Mir, S. Leonardi, A. Pistone, L. Mavilia, and G. Neri. *Sensors and Actuators B:*
459 *Chemical*. **2014**, 196, 413-420.
- 460 [24] A. A. Darwish, M. M. Fadlallah, A. Badawi, and A. A. Maarouf. *Applied Surface Science*. **2016**, 377,
461 9-16.
- 462 [25] A. Seif, and K. Azizi. *RSC advances*. **2016**, 6, (6), 5079-5088.
- 463 [26] A. Seif, and K. Azizi. *RSC advances*. **2016**, 6, (63), 58458-58468.
- 464 [27] M. D. Esrafil, and S. Asadollahi. *Journal of Molecular Graphics and Modelling*. **2018**, 85, 323-330.
- 465 [28] Y. Lin, and J. W. Connell. *Nanoscale*. **2012**, 4, (22), 6908-6939.
- 466 [29] V. Guerra, C. Wan, and T. McNally. *Progress in Materials Science*. **2019**, 100, 170-186.
- 467 [30] J.-l. Li, J.-h. Yin, T. Ji, Y. Feng, Y.-y. Liu, H. Zhao, Y.-p. Li, C.-c. Zhu, D. Yue, and B. Su. *Materials*
468 *Letters*. **2019**, 234, 74-78.

- 469 [31] X. Yang, Y. Guo, Y. Han, Y. Li, T. Ma, M. Chen, J. Kong, J. Zhu, and J. Gu. *Composites Part B: Engineering*. **2019**, 175, 107070.
- 470
- 471 [32] L. H. Li, and Y. Chen. *Advanced Functional Materials*. **2016**, 26, (16), 2594-2608.
- 472 [33] A. Falin, Q. Cai, E. J. Santos, D. Scullion, D. Qian, R. Zhang, Z. Yang, S. Huang, K. Watanabe, and T. Taniguchi. *Nature communications*. **2017**, 8, (1), 1-9.
- 473
- 474 [34] A. A. Peyghan, M. Noei, and S. Yourdkhani. *Superlattices and Microstructures*. **2013**, 59, 115-122.
- 475 [35] R. Omidirad, and K. Azizi. *Journal of Molecular Graphics and Modelling*. **2019**, 93, 107448.
- 476 [36] M. D. Esrafil, and N. Saeidi. *Applied Surface Science*. **2018**, 444, 584-589.
- 477 [37] M. D. Esrafil. *Chemical Physics Letters*. **2018**, 695, 131-137.
- 478 [38] M. D. Esrafil, P. Mousavian, and F. A. Rad. *Journal of Molecular Graphics and Modelling*. **2018**, 82, 101-107.
- 479
- 480 [39] R. Moladoust, M. D. Esrafil, A. Hosseinian, I. Alkorta, and E. Vessally. *Molecular Physics*. **2019**, 117, (5), 626-634.
- 481
- 482 [40] M. D. Esrafil. *Solid State Communications*. **2018**, 284, 35-39.
- 483 [41] M. D. Esrafil, S. Asadollahi, and S. Heydari. *Journal of Molecular Graphics and Modelling*. **2019**, 89, 41-49.
- 484
- 485 [42] H. Li, Z. Chen, X. Fang, and D. Tie. *Superlattices and Microstructures*. **2015**, 88, 371-376.
- 486 [43] S. Lin, J. Huang, and X. Gao. *Physical Chemistry Chemical Physics*. **2015**, 17, (34), 22097-22105.
- 487 [44] J. Azamat, A. Khataee, and S. W. Joo. *Journal of Molecular Structure*. **2016**, 1108, 144-149.
- 488 [45] M. D. Esrafil, N. Saeidi, and P. Nematollahi. *New Journal of Chemistry*. **2016**, 40, (9), 8024-8031.
- 489 [46] M. D. Esrafil, and N. Saeidi. *International Journal of Quantum Chemistry*. **2017**, 117, (24), e25450.
- 490 [47] Z. Ai, B. Chang, C. Xu, B. Huang, Y. Wu, X. Hao, and Y. Shao. *New Journal of Chemistry*. **2019**, 43, (22), 8613-8619.
- 491
- 492 [48] R. W. Dorn, M. J. Ryan, T.-H. Kim, T. W. Goh, A. Venkatesh, P. M. Heintz, L. Zhou, W. Huang, and A. J. Rossini. *Chemistry of Materials*. **2020**, 32, (7), 3109-3121.
- 493
- 494 [49] L. H. Li, T. Xing, Y. Chen, and R. Jones. *Advanced materials interfaces*. **2014**, 1, (8), 1300132.
- 495 [50] L. Fu, G. Chen, N. Jiang, J. Yu, C.-T. Lin, and A. Yu. *Journal of Materials Chemistry A*. **2016**, 4, (48), 19107-19115.
- 496
- 497 [51] W. Luo, Y. Wang, E. Hitz, Y. Lin, B. Yang, and L. Hu. *Advanced Functional Materials*. **2017**, 27, (31), 1701450.
- 498
- 499 [52] M. A. Raza, A. Nadeem, and M. T. Ilyas. "Corrosion Study of Boron Nitride Nanosheets Deposited on Copper Metal by Electrophoretic Deposition." pp. 681-685.
- 500
- 501 [53] P. Hohenberg, and W. Kohn. *Physical review*. **1964**, 136, (3B), B864.
- 502 [54] W. Kohn, and L. J. Sham. *Physical review*. **1965**, 140, (4A), A1133.
- 503 [55] J. A. Pople, P. M. Gill, and B. G. Johnson. *Chemical physics letters*. **1992**, 199, (6), 557-560.
- 504 [56] F. M. Bickelhaupt, and E. J. Baerends. *Reviews in computational chemistry*. **2000**, 15, 1-86.
- 505 [57] J. P. Perdew, K. Burke, and M. Ernzerhof. *Physical review letters*. **1996**, 77, (18), 3865.
- 506 [58] J. P. Perdew, M. Ernzerhof, and K. Burke. *The Journal of chemical physics*. **1996**, 105, (22), 9982-9985.
- 507
- 508 [59] C. Adamo, and V. Barone. *The Journal of chemical physics*. **1999**, 110, (13), 6158-6170.
- 509 [60] Y. Zhao, and D. G. Truhlar. *Theoretical Chemistry Accounts*. **2008**, 120, (1-3), 215-241.
- 510 [61] J.-D. Chai, and M. Head-Gordon. *Physical Chemistry Chemical Physics*. **2008**, 10, (44), 6615-6620.
- 511 [62] S. Grimme. *Journal of computational chemistry*. **2006**, 27, (15), 1787-1799.
- 512 [63] S. Grimme, J. Antony, S. Ehrlich, and H. Krieg. *The Journal of chemical physics*. **2010**, 132, (15), 154104.
- 513
- 514 [64] S. Grimme, S. Ehrlich, and L. Goerigk. *Journal of computational chemistry*. **2011**, 32, (7), 1456-1465.
- 515

516 [65] P. J. Hay. *The Journal of Chemical Physics*. **1977**, 66, (10), 4377-4384.
517 [66] R. Krishnan, J. S. Binkley, R. Seeger, and J. A. Pople. *The Journal of chemical physics*. **1980**, 72, (1),
518 650-654.
519 [67] M. J. Frisch, J. A. Pople, and J. S. Binkley. *The Journal of chemical physics*. **1984**, 80, (7), 3265-
520 3269.
521 [68] K. Raghavachari, and G. W. Trucks. *The Journal of chemical physics*. **1989**, 91, (2), 1062-1065.
522 [69] R. Binning Jr, and L. Curtiss. *Journal of Computational Chemistry*. **1990**, 11, (10), 1206-1216.
523 [70] M. P. McGrath, and L. Radom. *The Journal of chemical physics*. **1991**, 94, (1), 511-516.
524 [71] T. V. Russo, R. L. Martin, and P. J. Hay. *The Journal of chemical physics*. **1994**, 101, (9), 7729-7737.
525 [72] L. A. Curtiss, M. P. McGrath, J. P. Blaudeau, N. E. Davis, R. C. Binning Jr, and L. Radom. *The Journal*
526 *of Chemical Physics*. **1995**, 103, (14), 6104-6113.
527 [73] J.-P. Blaudeau, M. P. McGrath, L. A. Curtiss, and L. Radom. *The Journal of chemical physics*. **1997**,
528 107, (13), 5016-5021.
529 [74] L. Goerigk, and S. Grimme. *Physical Chemistry Chemical Physics*. **2011**, 13, (14), 6670-6688.
530 [75] H. Ibach, and H. Lüth. *Solid State and Physics, Springer, Berlin*. **1995**, 244.
531 [76] R. Dennington, T. A. Keith, and J. M. Millam. *Semichem Inc. Shawnee Mission KS*. **2016**,
532 [77] J. Baker. *Journal of computational chemistry*. **1987**, 8, (5), 563-574.
533 [78] M. Frisch, G. Trucks, H. Schlegel, G. Scuseria, M. Robb, J. Cheeseman, G. Scalmani, V. Barone, G.
534 Petersson, and H. Nakatsuji, "Gaussian 16," Gaussian, Inc. Wallingford, CT, 2016.
535 [79] T. Lu, and F. Chen. *Journal of computational chemistry*. **2012**, 33, (5), 580-592.
536 [80] N. M. O'boyle, A. L. Tenderholt, and K. M. Langner. *Journal of computational chemistry*. **2008**, 29,
537 (5), 839-845.
538 [81] S. F. Boys, and F. Bernardi. *Molecular Physics*. **1970**, 19, (4), 553-566.
539 [82] I. Alkorta, C. Trujillo, J. Elguero, and M. Solimannejad. *Computational and Theoretical Chemistry*.
540 **2011**, 967, (1), 147-151.
541 [83] F. Ma, Z.-R. Li, Z.-J. Zhou, D. Wu, Y. Li, Y.-F. Wang, and Z.-S. Li. *The Journal of Physical Chemistry C*.
542 **2010**, 114, (25), 11242-11247.
543 [84] J. Foresman, and E. Frisch. *Gaussian Inc., Pittsburgh, Pennsylvania*. **1996**,
544 [85] J.-L. Bredas. *Materials Horizons*. **2014**, 1, (1), 17-19.
545 [86] P. Geerlings, F. De Proft, and W. Langenaeker. *Chemical reviews*. **2003**, 103, (5), 1793-1874.
546 [87] S.-B. Liu. *Acta Physico-Chimica Sinica*. **2009**, 25, (3), 590-600.
547 [88] a. J. Foster, and F. Weinhold. *Journal of the American Chemical Society*. **1980**, 102, (24), 7211-
548 7218.
549 [89] F. Weinhold, and C. R. Landis. *Chemistry Education Research and Practice*. **2001**, 2, (2), 91-104.
550 [90] F. Weinhold, *Discovering chemistry with natural bond orbitals*: John Wiley & Sons, 2012.
551 [91] R. S. Mulliken. *The Journal of Chemical Physics*. **1955**, 23, (10), 1833-1840.
552 [92] I. Mayer. *Chemical Physics Letters*. **1983**, 97, (3), 270-274.
553 [93] A. J. Bridgeman, G. Cavigliasso, L. R. Ireland, and J. Rothery. *Journal of the Chemical Society, Dalton*
554 *Transactions*. **2001**, (14), 2095-2108.
555 [94] I. Mayer. *Chemical Physics Letters*. **2012**, 544, 83-86.
556 [95] K. B. Wiberg. *Tetrahedron*. **1968**, 24, (3), 1083-1096.
557 [96] O. V. Sizova, L. V. Skripnikov, and A. Y. Sokolov. *Journal of Molecular Structure: THEOCHEM*. **2008**,
558 870, (1-3), 1-9.
559 [97] C. F. Matta, "Hydrogen–Hydrogen Bonding: The Non-Electrostatic Limit of Closed-Shell Interaction
560 Between Two Hydro," *Hydrogen Bonding—New Insights*, pp. 337-375: Springer, 2006.
561 [98] H. J. Bohórquez, R. J. Boyd, and C. F. Matta. *The Journal of Physical Chemistry A*. **2011**, 115, (45),
562 12991-12997.

- 563 [99] S. J. Grabowski. *The Journal of Physical Chemistry A*. **2012**, 116, (7), 1838-1845.
- 564 [100] E. R. Johnson, S. Keinan, P. Mori-Sánchez, J. Contreras-García, A. J. Cohen, and W. Yang. *Journal*
565 *of the American Chemical Society*. **2010**, 132, (18), 6498-6506.
- 566 [101] J. Contreras-García, E. R. Johnson, S. Keinan, R. Chaudret, J.-P. Piquemal, D. N. Beratan, and W.
567 Yang. *Journal of chemical theory and computation*. **2011**, 7, (3), 625-632.

568