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Preprint Title Topographic signatures and manipulations of Fe atoms, CO molecules and NaCl islands on superconducting Pb(111)

Authors Carl Drechsel, Philipp D'Astolfo, Jung-Ching Liu, Thilo Glatzel, Rémy Pawlak and Ernst Meyer

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ORCID® IDs Rémy Pawlak - <https://orcid.org/0000-0001-8295-7241>

1 **Topographic signatures and manipulations of Fe atoms, CO molecules** 2 **and NaCl islands on superconducting Pb(111)**

3 Carl Drechsel¹, Philipp D' Astolfo¹, Jung-Ching Liu¹, Thilo Glatzel¹, Rémy Pawlak*¹ and Ernst
4 Meyer*¹

5 Address: ¹Department of Physics, Universität Basel, Klingelbergstrasse 82, 4056 Basel, Switzer-
6 land

7 Email: Rémy Pawlak - remy.pawlak@unibas.ch; Ernst Meyer - ernst.meyer@unibas.ch

8 * Corresponding author

9 **Abstract**

10 **Background:** Topological superconductivity emerging in one- or two-dimensional hybrid ma-
11 terials is predicted as a key ingredient for quantum computing. However, not only the design of
12 complex heterostructures is primordial for future applications but also the characterization of their
13 electronic and structural properties at the atomic scale using the most advanced scanning probe mi-
14 croscopy techniques with functionalized tips.

15 **Results:** We report on the topographic signatures observed by scanning tunneling microscopy
16 (STM) of carbon monoxide (CO) molecules, iron (Fe) atoms and sodium chloride (NaCl) islands
17 deposited on superconducting Pb(111). For the CO adsorption a comparison with the Pb(110) sub-
18 strate is demonstrated. We show a general propensity of these adsorbates to diffuse at low tempera-
19 ture under gentle scanning conditions.

20 **Conclusion:** Our findings provide new insights into high-resolution probe microscopy imaging
21 with terminated tips, decoupling atoms and molecules by NaCl islands or tip-induced lateral ma-
22 nipulation of iron atoms on top of the prototypical Pb(111) superconducting surface.

23 **Keywords**

24 Superconductivity, NaCl, CO, scanning tunneling microscopy, lateral manipulation

25 Introduction

26 The most exciting manifestation of topological superconductivity [1-3] is the Majorana zero mode
27 (MZM), which has attracted a tremendous interest due to its non-Abelian quantum exchange statis-
28 tics proposed as a key ingredient for topological quantum computing [4-6]. Topological supercon-
29 ductivity can intrinsically arise in the bulk of certain materials [7] or can be engineered at the in-
30 terface between two materials, exhibiting particle-hole symmetry and spin-orbit interaction [8].

31 Among the most promising platforms to realize MZMs are semiconducting nanowires with large
32 spin-orbit coupling [9-12] or atomic chains [13-18] in proximity to an *s*-wave superconductor. The
33 realization of MZMs in two-dimensions has been also observed in vortex cores on a proximitized
34 topological insulator surface [19,20], in iron-based superconductors [7,21,22] or hybrid van der
35 Waals heterostructures [23]. The fingerprint for MZMs in conductance measurements through the
36 nanowire or in scanning tunneling spectroscopy (STS) is a zero-bias conductance peak occurring at
37 boundaries and defects. Unfortunately, other structural peculiarities can also mimic such zero-bias
38 anomalies, which eventually leads to severe misinterpretations. Therefore, the latest advances in
39 scanning tunneling microscopy (STM) and atomic force microscopy (AFM) are required to accu-
40 rately disentangle structural and electronic properties of these MZMs platforms.

41 The development of functionalized tips, obtained by picking up a single molecule from a surface,
42 has been an important milestone for low temperature STM/AFM techniques since the CO tip nowa-
43 days enables systematic high-resolution measurements of surfaces, molecules and atoms [24-26].
44 STM/AFM indeed allows controlled repositioning of adsorbates, both in vertical and lateral direc-
45 tions. Atoms and molecules can be pushed or pulled across a surface [27-29], but also picked up
46 and dropped with the probing tip [30-33]. Lateral repositioning of adatoms is also an promising
47 asset as it could serve as unique opportunity to design atomic structures with novel electronic prop-
48 erties [27,34,35].

49 It is however astonishing, that most recent advances in manipulation experiments or contrast en-
50 hancement with functionalized tips are hitherto at their infancy, when studying a superconducting
51 surface by STM/AFM. Although the earliest proposal for observing MZMs suggested to repositi-

52 tion Fe adatoms one-by-one with an STM tip in an one-dimensional fashion on an *s*-wave super-
53 conductor [10], this strategy has been primarily postponed in favor of self-assembly processes on
54 Pb(110) surfaces [14,15,36,37]. Only recently, the successful manipulation of tens of Fe atoms has
55 been reported on superconducting Re(0001) [16] and Ta(100)-O surfaces [38]. Despite being well-
56 established on many noble metals, the use of CO-terminated tips also remains quite scarce in the
57 literature [28], which severely limits the use of AFM as imaging tool on superconductors.
58 Recently, Heinrich *et al.* have demonstrated the possibility to tune the magnetic anisotropy of a sin-
59 gle porphyrin molecule by perturbing its ligand field with the STM probe [39,40]. These results
60 not only suggest the importance of future manipulations experiments, but also shed new lights into
61 the potential of decoupling atoms and molecules electronically from the underlying superconduc-
62 tors. With this prospect, we emphasize that in addition to tip manipulations the use of alkali-halide
63 islands, adsorbed on a superconducting surface and acting as a buffer layer, is another interesting
64 field for research on topological superconductors [41-44].
65 In this work, we report on the topographic features of adsorbed CO molecules, NaCl layers and
66 iron adatoms on a superconducting Pb(111) surface, investigated with STM at 4.8 K. We show that
67 CO molecules on Pb(111) are hardly visible in STM images due to their high diffusion induced by
68 the tip even at low temperature. This differs slightly from the adsorption on Pb(110), which has
69 also been performed. In contrast, NaCl islands and single Fe atoms are more stable. Nevertheless,
70 a general propensity for a tip-induced displacement of these adsorbates on the Pb(111) surface can
71 be fulfilled. We believe that our results constitute an important step for future experiments to per-
72 form high-resolution STM/AFM imaging with CO-terminated tips or the electronic decoupling of
73 atoms and molecules from the prototypical Pb(111) superconducting surface.

74 **Experimental**

75 **Sample preparation**

76 The Pb(111) single crystal, purchased from Mateck GmbH, was cleaned by several sputtering and
77 annealing cycles in ultra-high vacuum (UHV). CO dosing on the cold substrate was done in the

78 microscope chamber by increasing the pressure via a leak valve up to $p \approx 1 \times 10^{-7}$ mbar for one
79 minute. This leads to a surface coverage of about 0.1-0.3 monolayer, as we readily observed on no-
80 ble metals such as Cu, Ag or Au [33,45]. Iron adatoms were evaporated in the microscope head on
81 the substrate at a temperature below 15 K. NaCl was evaporated from a quartz crucible on samples
82 kept at room temperature in the preparation chamber.

83 **Low-temperature scanning tunneling microscope**

84 The experiments were performed using a low-temperature STM/AFM microscope ($T = 4.8$ K)
85 from Omicron GmbH in UHV ($p \approx 1 \times 10^{-10}$ mbar) operated with Nanonis RC5 electronics. The
86 sensor is a tuning fork sensor in a qPlus design [46] operated in the frequency-modulation mode
87 (resonance frequency $f_0 \approx 25$ kHz, spring constant $k \approx 1800$ N/m, quality factor $Q \approx 14000$, and
88 oscillation amplitude $A \approx 0.5$ Å). The tip mounted to the qPlus sensor consists of a $25 \mu\text{m}$ -thick
89 PtIr wire, shortened and sharpened with a focused ion beam. A clean and sharp Pb tip was then
90 prepared at low temperature by repeated indentations into the surface. STM images were acquired
91 in constant-current mode with the bias voltage applied to the tip. All experimental data were anal-
92 ysed by using Gwyddion [47].

93 **Results and Discussion**

94 **CO adsorption on Pb(111) and Pb(110)**

95 Figure 1 shows STM images of CO molecules adsorbed on Pb(111). With a lattice parameter of
96 $a_{\text{Pb}} = 4.95$ Å, the height of mono-atomic steps of the Pb(111) surface is expected to be $h_{\text{Pb}} =$
97 $a_{\text{Pb}}/2\sqrt{3} = 1.4$ Å. Experimentally, a pristine Pb(111) sample (Fig. 1a) shows after sputtering and
98 annealing cycles typically steps of about $2 \times h_{\text{Pb}} = 2.7$ Å, which thus corresponds to diatomic
99 steps. We assume that this peculiar step height distribution results from quantum size effect of the
100 Pb(111) surface [48]. On the terraces, hexagonal dark spots are visible by STM, whose diameters
101 vary between 1.5 nm and 5 nm with an apparent depression of 0.14 Å. They result from the inter-
102 ference of bulk electrons with trapped subsurface Ar gas bubbles after sputtering [49,50].

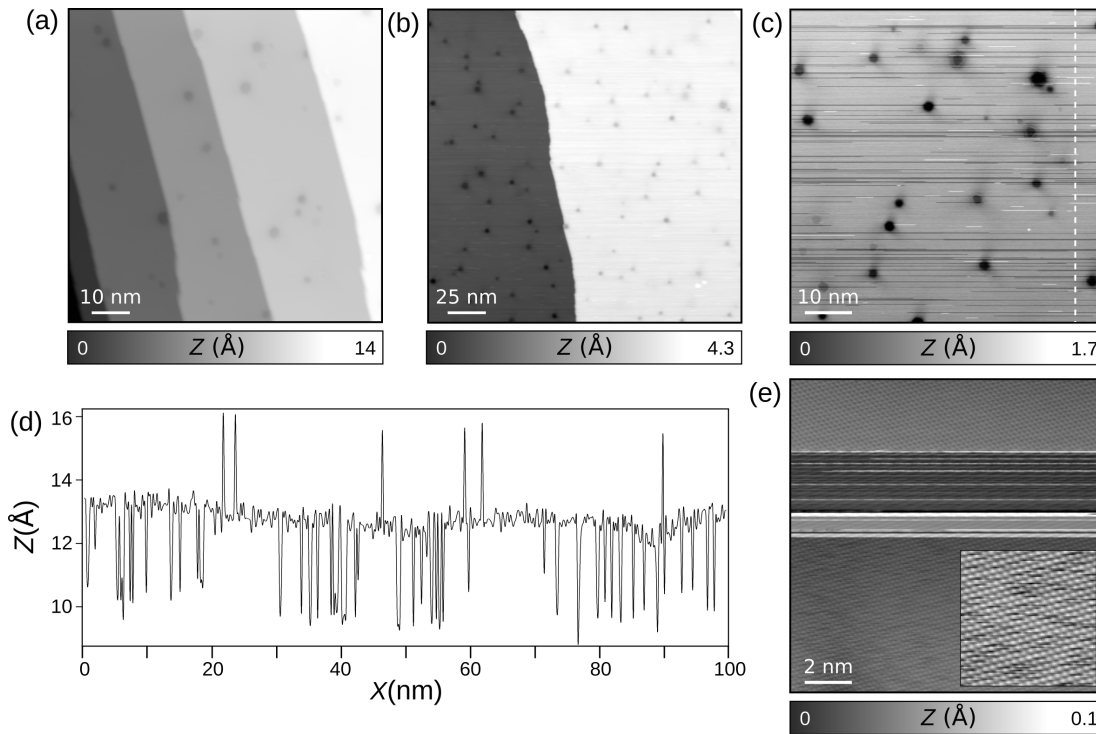


Figure 1: Carbon monoxide (CO) molecules adsorbed on Pb(111). (a) STM overview image of pristine Pb(111) ($V_t = -0.1$ V, $I_t = 1$ pA). (b) STM image after CO deposition. The estimated coverage is below 0.2 monolayer. (c) Close-up STM topography of CO molecules diffusing on the surface during scanning ($V_t = -0.1$ V, $I_t = 40$ pA). (d) Profile taken along the dashed white line of (c) showing spontaneous CO displacement under tip action. (e) Enhanced STM resolution (inset: more detailed image of the Pb(111) surface) resulting from the termination of the tip by a CO molecule ($V_t = -0.2$ V, $I_t = 1$ pA).

103 After CO dosing in the microscope chamber (see Sample preparation), a coverage of about 0.1-
 104 0.2 monolayer is expected to adsorb on the metal surface, as observed on different noble met-
 105 als [33,45]. Figures 1b and c show STM topographic images after such process. While the sur-
 106 face topography remains unchanged in comparison to Figure 1a, numerous scan instabilities are
 107 now present which we attribute to CO molecules diffusing under gentle scan conditions (tunnel-
 108 ing resistance of 200 G Ω). The STM profile (Figure 1d) taken along the white dashed line of Fig-
 109 ure 1c shows several stochastic jumps, which we interpret as tip-induced displacements of single
 110 CO molecules [51-53]. We emphasize that the change of various scan parameters as well as tip in-
 111 dentations into the clean Pb surface were conducted to avoid such instabilities without noticeable
 112 improvements. Nevertheless, an unintentional CO-tip termination could be achieved as shown by
 113 the enhancement of the STM resolution in Figure 1e. In comparison to vertical manipulations of

114 CO on noble metals, we emphasize that CO-terminated tips on Pb(111) are much less stable, which
 115 severely limits the use of CO-terminated STM/AFM imaging on Pb(111). It should be noted that
 116 also other tip terminations are possible (such as with Xe), which we plan to explore in future work.

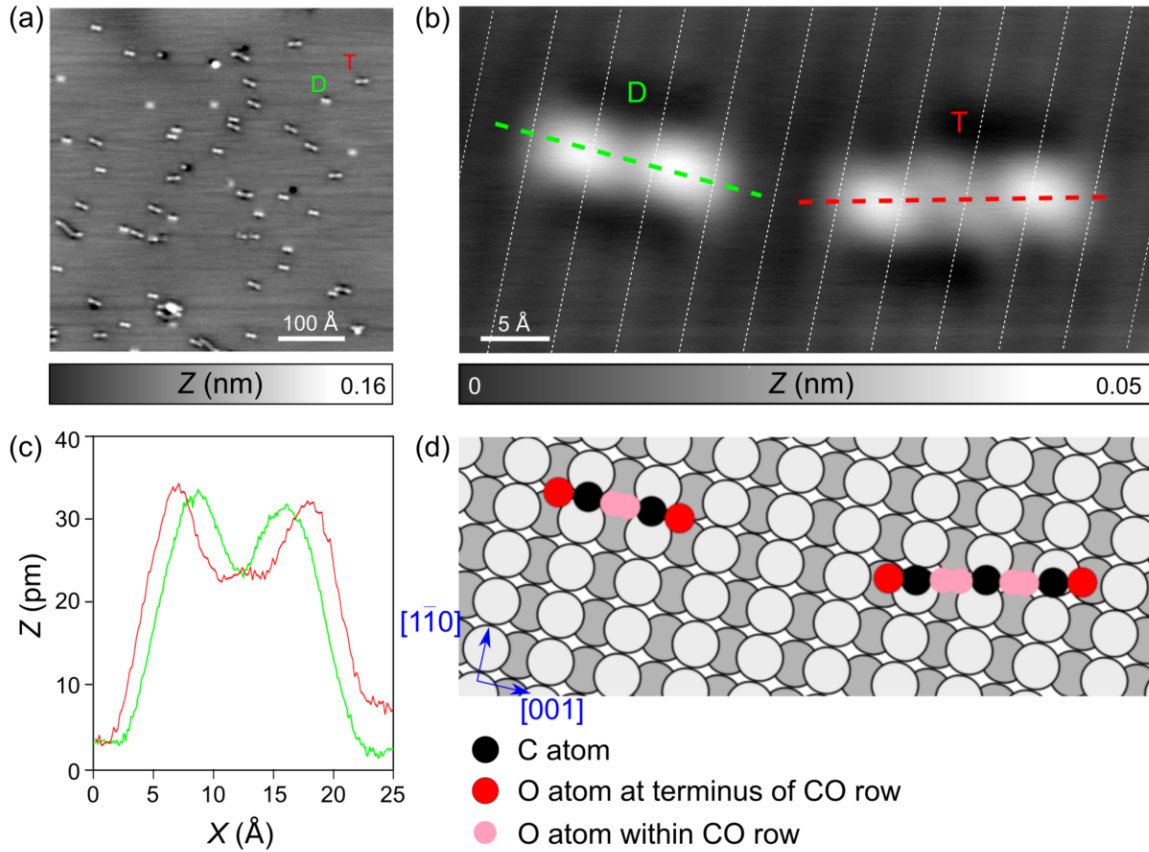


Figure 2: Adsorption of carbon monoxide (CO) molecules on Pb(110). (a) STM overview image of Pb(110) after adsorption of CO molecules (D: dimer, T: trimer, $V_t = -0.5$ V, $I_t = 0.5$ pA). (b) Closed-up STM image of a CO dimer and trimer ($V_t = -0.1$ V, $I_t = 1$ pA). (c) Profiles taken along the dashed lines of (b). (d) Sphere model of CO adsorbed on Pb(110). The CO molecules are standing up with protruding and tilted oxygen atoms (red, if tilted to chain terminus; pink, if tilted to center of CO chain), white and dark gray spheres refers to the topmost and downmost Pb atoms of the Pb(110) reconstruction.

117 Similar CO adsorptions were also conducted on Pb(110) (Figure 2a). There, most CO molecules
 118 appear in STM images as linear aggregates of different lengths, aligned nearly perpendicular to the
 119 $[1\bar{1}0]$ row direction of Pb(110). The dimer-like protrusions ((D) in Figure 2b) exhibit a length of
 120 ≈ 7 Å between maxima (Figure 2c), corresponding to the distance of $a_{Pb} = 4.95$ Å between two
 121 Pb(110) rows (dashed lines in Figure 2b). The additional length of ≈ 2 Å might be related to the

122 tilting of the CO molecules under the scanning tip as well as the tip convolution during imaging.
123 The trimeric protrusion (T) is rotated by about 16° compared to the $[1\bar{1}0]$ rows. Its length of \approx
124 11 \AA corresponds to about three Pb(110) atomic rows, the additional length of $\approx 1 \text{ \AA}$ is again im-
125 puted to tilted CO during tip scanning. Last, these protrusions have a slight apparent depression
126 around them, which might be related to a locally induced strain of the Pb lattice. While the D fea-
127 tures are all aligned in the same direction, the orientation of the T features differs slightly. Both
128 features have an apparent height of $\approx 0.3 \text{ \AA}$, as extracted from the profile of Figure 2b, displayed in
129 Figure 2c.

130 Overall, the CO adsorption on Pb(110) shows very strong similarities with adsorbed CO on
131 Cu(110)-(2 \times 1)O as reported in references [54,55]. There, self-assembled CO molecules chemisorb
132 with the C atom on top of Cu-O rows. According to DFT calculations, they lift the host Cu atom
133 by 1 \AA and the entire Cu-CO unit is tilted by $\approx \pm 45^\circ$. For measurements at 77 K, the two tilting
134 configurations are supposed to convert rapidly, while at 4.5 K they are stable in one tilted configu-
135 ration. At low coverages, CO adsorbed mostly as monomers and dimers, only occasionally trimers
136 or even longer configurations were observed. Consecutive measurements showed however, that cer-
137 tain monomers formed dimers after some time. In STM the tilted CO molecules showed for each
138 feature at the edges higher contrast as the interior molecules, which was explained by the different
139 chemical environment. With this model, supported by DFT, reference [54] explained the formation
140 of CO rows by dipole-dipole interactions, caused by the displacement of the Cu atom. They can be
141 repulsive for vertical CO molecules [56,57], but attractive for tilted ones [54].

142 If we transfer this model to our measurements, the adsorption of the CO molecules might take
143 place on top of the $[1\bar{1}0]$ rows of the Pb(110) surface, as shown by the model in Figure 2d. For
144 the dimer (D), the C atom is probably bonded to the Pb at the bridge sites of $[1\bar{1}0]$ rows. The CO
145 molecule is tilted similarly as on the Cu(110)-(2 \times 1)O surface [54,55] and appears in STM above
146 the trenches of the Pb(110) surface. For the trimer (T), the mutual interaction of the interior CO
147 molecules might cause a slight mismatch with the Pb(110) layer, which explains the small devi-

148 ation from the perpendicular alignment of the dimers. For longer CO aggregates, this deviation
 149 becomes even more apparent (see Figure 2a).
 150 The reason, why CO can be observed with STM on Pb(110) and not on Pb(111) might be the fact,
 151 that in the $[1\bar{1}0]$ rows of Pb(110) the attractive dipole-dipole interaction is initiated by a slight lift
 152 of the Pb atoms, which is not possible in the dense Pb(111) surface. It is also remarkable that the
 153 instability of the tilting angle of CO on Pb(110) at 4.5 K is comparable to that of CO on Cu(110)-
 154 $(2\times 1)O$ at 77 K.

155 Growth of NaCl islands on Pb(111)

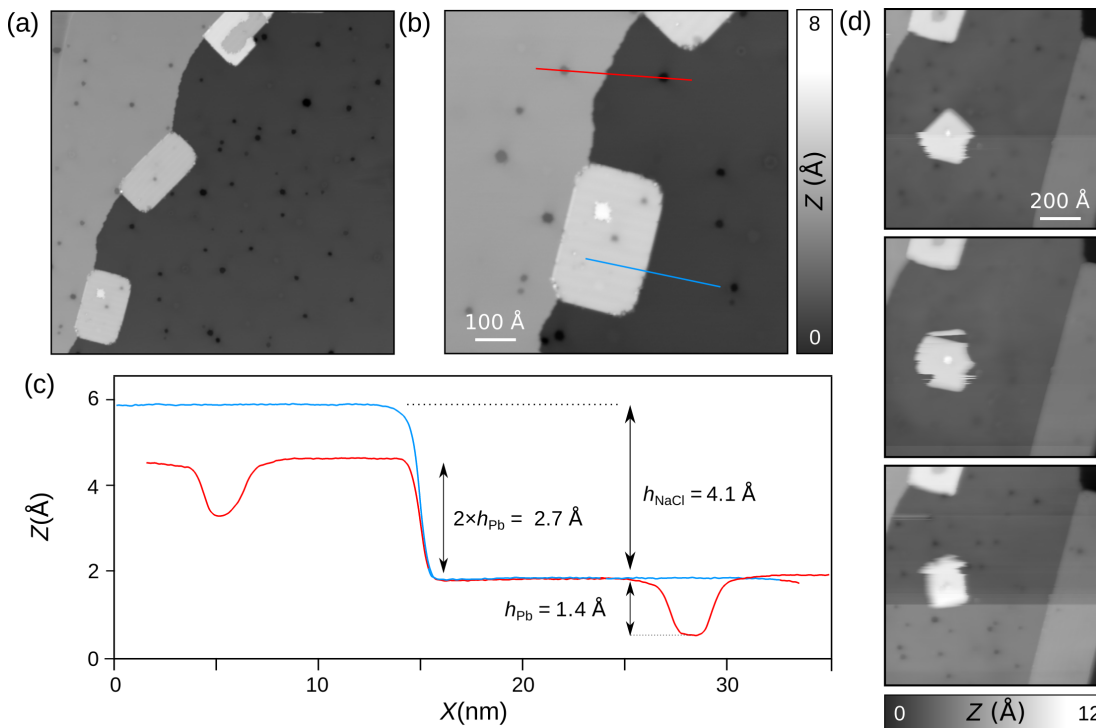


Figure 3: Adsorption of NaCl on Pb(111). **(a-b)** STM overview image of Pb(111) with quadratic NaCl islands adsorbed at step edges ($V_t = -0.4$ V, $I_t = 1$ pA). **(c)** Height profile extracted along the red and blue lines of **(b)**. **(d)** Series of STM image showing the tip-induced rotation of a NaCl island ($V_t = -0.4$ V, $I_t = 40$ pA).

156 We next investigated the adsorption of NaCl on Pb(111) (Figure 3). Upon sublimation from a
 157 quartz crucible on a Pb(111) surface, which is kept at room temperature, NaCl forms without any
 158 post-annealing rectangular islands with round shaped corners attached to Pb step edges (Figures 3a
 159 and b). According to the profile, shown in Figure 3c, which is extracted along the red and blue

160 lines of Figure 1b, the step heights are equal to $h_{NaCl} = 4.1 \text{ \AA}$. This corresponds to a NaCl bilayer
161 and is in agreement with the reported growth of NaCl islands on Cu(111) [58]. Occasionally, even
162 a trilayer phase appears within the NaCl bilayer (Figures 3a and b). Note also that dark protrusions
163 originating from trapped Ar atoms are still visible through the NaCl island by STM. Figure 3d
164 shows series of consecutive STM images of a NaCl island adsorbed on a terrace. Upon scanning
165 with a tunneling resistance of about $10 \text{ G}\Omega$, the entire island rotates under the tip action around a
166 trilayer signature as pinning center. This is in contrast to those NaCl islands, which are pinned to
167 step edges. They remain always stable at $T = 4.8 \text{ K}$, independent of the scanning conditions. As
168 is, these islands exhibit characteristics similar to the ones on conventional metals [41,43,44] and
169 thus they are likely adequate for the electronic decoupling of single atoms or molecules from the
170 superconducting Pb(111).

171 **Single Fe atoms on Pb(111) and their lateral manipulations**

172 Figure 4 shows the deposition and controlled lateral manipulation of Fe adatoms on Pb(111). Upon
173 deposition of Fe atoms on Pb(111) (kept below 15 K), several circular protrusions of different sizes
174 and heights are observed by STM (Figure 4a). Their lateral sizes range from 0.3 to 1.5 \AA , whereas
175 their heights exhibit values of 0.4 , 1.2 and 1.7 \AA . Although no atomic resolution of these aggre-
176 gates has been obtained, we interpret the variation of heights as a fingerprint for a Fe monomer,
177 dimer and trimer, respectively (denoted as Fe_1 , Fe_2 and Fe_3 in the following).

178 To confirm this assumption, we laterally manipulated single Fe adatoms with the STM tip [59,60]
179 to intentionally form dimers and trimers and measure their apparent STM heights. To do so, the
180 STM tip was positioned above a single Fe atom. The resistance of the STM junction was then
181 decreased from about $50 \text{ G}\Omega$ (imaging) to $3 \text{ G}\Omega$ (manipulation) in order to trap the Fe atom in
182 the STM junction [61]. Upon lateral tip displacements with a velocity of about $500 \text{ pm}\cdot\text{s}^{-1}$, the
183 trapped Fe atom is successfully displaced over the surface. During this process, a so-called "atom
184 manipulation image" [34] can be obtained from such dragging of the Fe atom over Pb(111) (Fig-
185 ure 4b). The geometric features resemble typical patterns observed in friction force microscopy

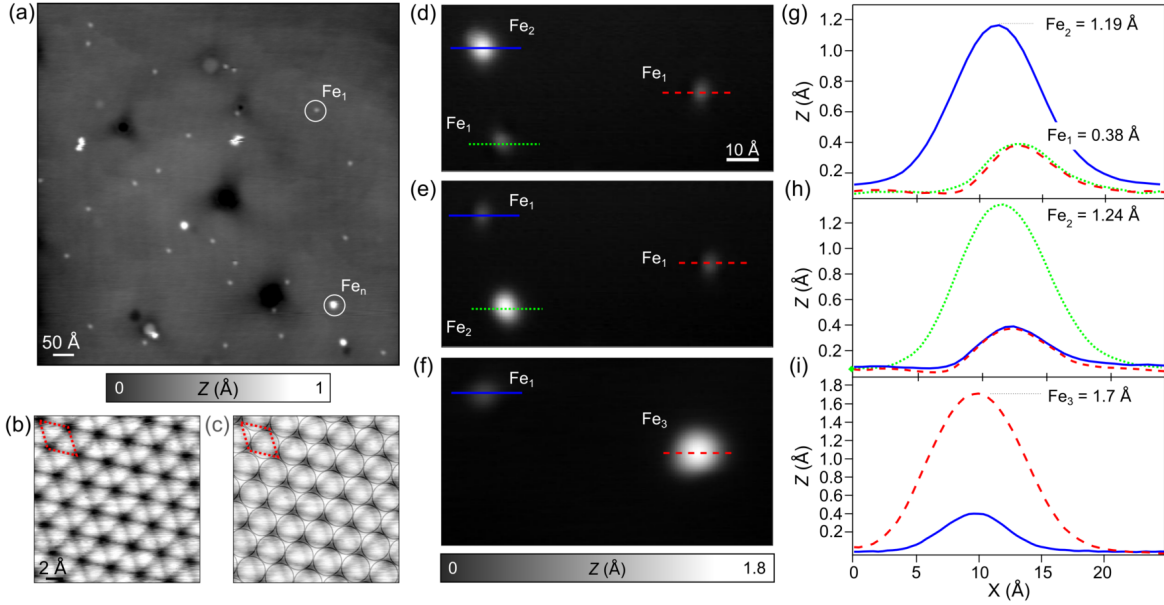


Figure 4: Fe adatoms on Pb(111) and their lateral manipulations. (a) STM overview image of Pb(111) after deposition of Fe adatoms ($V_t = -700$ mV, $I_t = 5$ pA). Fe_1 and Fe_n correspond to single Fe adatom and clusters of n adatoms, respectively ($V_t = -15$ mV, $I_t = 5$ pA). (b) Topographic STM image during the manipulation of a single Fe atoms trapped in the STM junction. (c) Models of the Pb(111) corresponding to the STM image of (b). The red dashed parallelogram refers to the Pb(111) lattice. (d-f) Series of STM images of Fe adatoms and their successive lateral manipulations with the STM tip (marked by arrows in (d) and (e)). In (f), the STM image shows the formation of a Fe-trimer Fe_3 by successive tip manipulations (Imaging conditions, $V_t = -30$ mV, $I_t = 60$ pA). (g-i) Apparent STM heights extracted from images (d-f) enabling one to distinguish from their topographic signatures Fe_1 , Fe_2 and Fe_3 , respectively.

186 (FFM) [28,29], since the trapped Fe atom senses the surface potential in analogy to the probing
 187 tip of FFM. For clarity, we overlay the Pb(111) surface lattice on top of the image in Figure 4c.
 188 The darkest features are spaced by 0.35 \AA in agreement with the lattice parameters of Pb(111) and
 189 likely correspond to hollow sites, where the adatom preferentially is located.
 190 Using this method, we transferred single atoms between different Fe clusters. In Figure 4d, two Fe
 191 single atoms (Fe_1) and an assumed dimer (Fe_2) are displayed. Figure 4g shows the corresponding
 192 apparent STM heights, which can be extracted from the lain, dotted and dashed lines of Figure 4d.
 193 Thus, we infer the heights of Fe_1 and Fe_2 aggregates to be $h_1 \approx 0.4 \text{ \AA}$ and $h_2 \approx 1.2 \text{ \AA}$, respectively.
 194 As a verification, we then conducted the transfer of a single Fe atom (shown by the arrow in Fig-
 195 ure 4d) from the Fe_2 cluster to one surrounding Fe_1 in order to form a new dimer. The result of

196 such manipulation is shown in Figure 4e. Despite the exchange of Fe atoms by tip manipulation,
197 the apparent height of Fe₁ and Fe₂ remains identical as demonstrated by the STM profile of Fig-
198 ure 4h.

199 Finally, we brought by two successive tip manipulations the atoms of Fe₂ in Figure 4e to a third
200 single atom (see arrow). The resulting image (Figure 4f) reveals the formation of a Fe-trimer (Fe₃).
201 Compared to the heights of Fe₁ and Fe₂, the Fe₃ height is about $h_3 = 1.7 \text{ \AA}$. This evolution of STM
202 apparent heights as a function of number of atoms in small Fe clusters is in good agreement with a
203 similar study of Fe clusters on Cu(111) [62].

204 **Conclusion**

205 Our results report on the systematic characterization by STM of the adsorption of carbon monox-
206 ide (CO), sodium chloride (NaCl) and iron adatoms (Fe) on the superconducting Pb(111) surface
207 at low temperature (4.7 K). We show a surprising absence of STM topographic signatures of CO
208 molecules on Pb(111), which we impute to their high propensity of diffusing under gentle scan-
209 ning conditions. In contrast, CO molecules become apparent by STM on Pb(110), since they ini-
210 tiate attractive dipole-dipole interactions, which support the formation of linear aggregates. Fur-
211 thermore, we show that deposition of NaCl on Pb(111) leads to bilayer islands similar to litera-
212 ture data. Lastly, cold-temperature deposition ($\leq 15 \text{ K}$) of Fe on Pb(111) leads to the adsorption
213 of adatoms and small Fe clusters. Using tip-induced lateral manipulations, we demonstrate the ex-
214 change of Fe single atoms between these clusters and characterize the variation of apparent STM
215 height of each cluster as a function of the number of atoms. Overall, our findings provide new in-
216 sights into high-resolution STM/AFM imaging with functionalized tips, decoupling of atoms or
217 molecules and tip-induced lateral manipulation of Fe atoms above the prototypical Pb(111) super-
218 conducting surface.

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